

## ZnO–CdSe Nanoparticle Clusters as Directional Photoemitters with Tunable Wavelength

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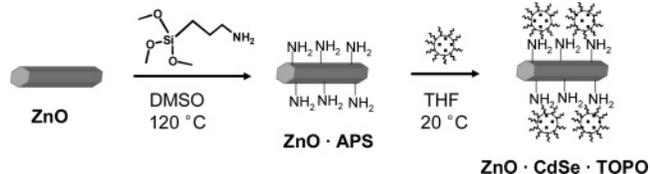
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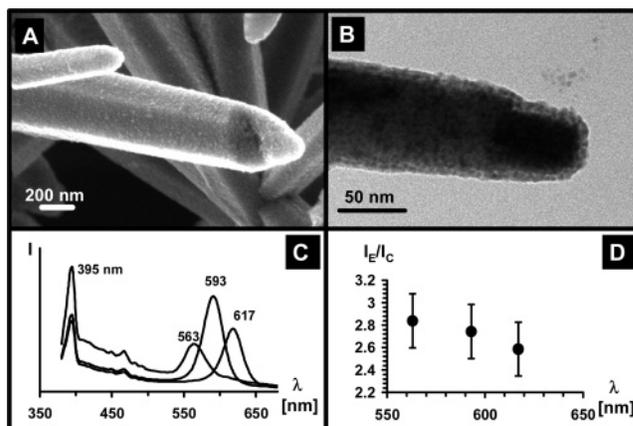
Because of their tailorable optical properties, semiconductor nanocrystals are considered key materials for applications as optical probes,<sup>1</sup> and as components in photochemical cells,<sup>2</sup> light emitting devices,<sup>3</sup> and lasers.<sup>4–6</sup> While emission and absorption energies of the nanoparticles can be adjusted with the particle sizes,<sup>7,8</sup> the polarization of the emission can be controlled with the particle shape.<sup>9,10</sup> For ZnO nanowires, it has recently been shown that the exciton emission occurs preferentially in the direction of the longer axis of the nanocrystal.<sup>11</sup> This waveguiding effect is a result of total internal reflection of the emitted light at the microcrystal–air interface.<sup>11,12</sup> We now found that the waveguiding effect of ZnO microcrystals can be exploited for directing the emission of CdSe nanoparticles by covalently linking the latter to the surface of a ZnO microcrystal rod. The resulting nanoparticle clusters exhibit directional photoluminescence at a wavelength that can be separately adjusted with the CdSe nanocrystal size. The ability to separately control the emission direction and wavelength of semiconducting nanoparticles should be of interest for applications of such structures in displays with enhanced contrast, in microlasers, and in photonics devices.

Unidirectional light emitters were synthesized according to Scheme 1 from rod-shaped ZnO microcrystals<sup>13</sup> and TOPO coated CdSe/CdS core/shell nanoparticles.<sup>16</sup>

### Scheme 1



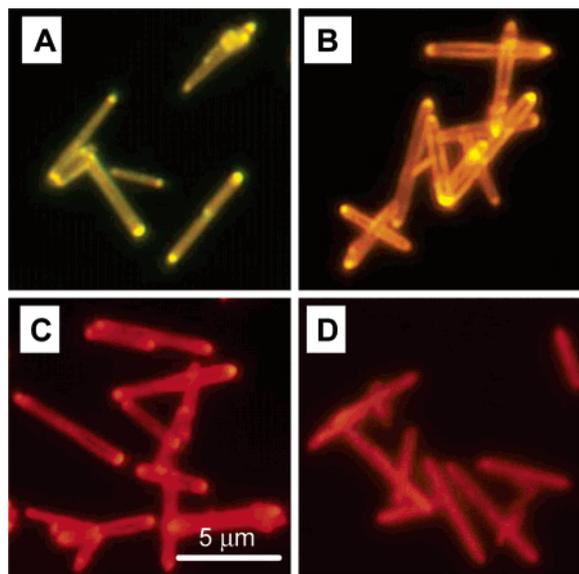
ZnO was chosen as a structural and optical support because of its high refractive index of 2.01,<sup>14</sup> and because as a wide band gap semiconductor (3.2 eV) it does not absorb visible light. Reaction of the ZnO microrods with 3-aminopropyltrimethoxysilane (APS) produced APS-terminated ZnO rods, which, in a subsequent reaction with trioctylphosphineoxide (TOPO)-stabilized CdSe/CdS core-shell nanoparticles in THF, underwent coupling to yield ZnO–CdSe clusters. In this reaction the primary amine group from APS displaces the weakly bonded TOPO from the CdS surface. SEM and TEM micrographs (Figure 1a,b) show that the clusters are composed of ZnO microrods that are coated with a dense ~10-nm-thick monolayer of CdSe particles. The sizes of the clusters ( $6.85 \pm 1.73 \mu\text{m}$  long and  $0.79 \pm 0.13 \mu\text{m}$  wide) are mainly determined by the sizes of the ZnO rods, which can be controlled by the conditions of ZnO synthesis. According to energy dispersive spectroscopy, the chemical composition (in mass %) of the clusters is  $1.61 \pm 0.21$  (Se),  $2.23 \pm 0.25$  (Cd),  $68.78 \pm 0.47$  (Zn),  $20.49 \pm 0.70$  (O),  $6.02 \pm 0.61$  (C), and  $0.89 \pm 0.07$  (Si). The relatively high carbon content indicates that the CdSe/CdS particles are partially coated with TOPO.



**Figure 1.** (A,B) SEM and TEM images of ZnO–CdSe clusters. A monolayer of CdSe particles is noticeable. (C) Fluorescence spectra of ZnO–CdSe clusters in chloroform ( $\lambda_{\text{ex}} = 350 \text{ nm}$ ). (D) Plot of directional emission ratio versus emission wavelength.

Despite a relatively low calculated CdSe volume fraction (2.73%), the clusters strongly emit in the visible after excitation with visible light  $< 550 \text{ nm}$  (Figure 1c). Depending on the size of the CdSe nanocrystals (2.1–4.0 nm), the emission can be adjusted from yellow (563 nm) and orange (593 nm) to red (617 nm). When excited in the UV, the fluorescence spectrum of the particles also contains an emission feature at 395 nm corresponding to the band gap emissions of ZnO.<sup>11</sup> No defect emission of ZnO (normally at 510 nm) is observed. The CdSe emission quantum yield of the clusters is comparable to that of the free CdSe quantum dots in chloroform. The fact that there is little interaction between the two semiconductors might be due to the spatial separation between ZnO and CdSe ( $\sim 1.0 \text{ nm}$ ), which is enforced by the APS linker.

Wide field fluorescence micrographs of ZnO–CdSe clusters emitting yellow, orange, and red are presented in Figure 2a–c. All clusters appear dark in the center and light at the ends, which indicates that the emission occurs predominantly at the rod ends. This effect is strongest for clusters that are not in direct contact with the Si substrate (i.e., for clusters that are supported by other clusters). The mechanism for the observed directional light emission probably involves trapping of light from the photoexcited CdSe sites in the ZnO rods. Inside the optically dense ZnO rods the light is guided by total internal reflection at the ZnO–air interface. Using the indices of refraction of 2.01 and 1.00 for ZnO and air, respectively, Snell’s law predicts  $30^\circ$  as the critical angle for total internal reflection. When the clusters are immersed in a medium with higher index of refraction (e.g., microscopy oil with  $n = 1.59$ ), the directional emission is greatly diminished (Figure 2d). This confirms internal reflection as the mechanism for uniaxial light emission in these structures. To correlate the optical anisotropy of the clusters with their structural features, it is useful to calculate a ratio  $I_E/I_C$  from the emission intensities of rod ends ( $I_E$ ) and rod



**Figure 2.** Fluorescence micrographs of ZnO–CdSe clusters on a Si wafer in air (A–C) and in hydrocarbon oil with  $n = 1.59$  (D).

centers ( $I_C$ ), respectively, and to compare this “directional emission” ratio for different microrods. Surprisingly, the analysis (Supporting Information) shows that  $I_E/I_C$  (mean value  $2.72 \pm 0.45$ ) does not depend on either the thicknesses ( $0.48\text{--}1.06\ \mu\text{m}$ ) or the lengths ( $3.2\text{--}10.2\ \mu\text{m}$ ) of the clusters. This suggests that waveguiding in these structures is limited by scattering losses at the cluster surfaces. These losses are significant because of the rough surface structure of the clusters and because of the great refractive index ( $n = 2.45$ ) of the CdSe/CdS nanoparticles. As Figure 1d shows, there is also a weak inverse dependence of the directional emission ratio on the emission wavelengths ( $\lambda_{em}$ ) of the microstructures. This indicates that, as  $\lambda_{em}$  becomes larger, the light emitted by the CdSe nanocrystals is increasingly coupled to the Eigenmodes of the hexagonal ZnO rods, which act as optical cavities. The transition from multimode ( $W \gg 1$ ) to single mode ( $W \ll 1$ ) waveguiding is also expected from the waveguide parameter  $W = (2\pi\delta/\lambda_{em})(n_{ZnO}^2 - n_{air}^2)$  for these structures (see comments in Supporting Information), which lies between 3.49 and 3.67 (for mean radius  $\delta = 395$  nm, refractive index  $n = 2.01$ , emission wavelength  $\lambda_{em} = 560\text{--}620$  nm).<sup>15</sup> In terms of their ability to guide light of variable wavelengths, the ZnO–CdSe clusters thus represent the lower size limit.

In conclusion, we have demonstrated that directional emission of CdSe quantum dots can be achieved by chemical linkage of CdSe nanocrystals to ZnO microrods, which act as optical waveguides. In terms of their anisotropic optical properties, the ZnO–CdSe clusters resemble ZnO nanowires, for which a similar waveguiding mechanism has been postulated. The difference is that the light

emitted by ZnO–CdSe is generated by the CdSe nanoparticles and not by the ZnO rods, and that by changing the CdSe nanocrystal size, the emission wavelength of these structures can be adjusted. The described clusters are different from CdSe nanorods<sup>9</sup> in that they lack a polarization of the emitted light. In the latter structures, the emission is polarized due to a preferred orientation of the oscillating electrical dipole.

Our results suggest that the emission of other semiconductor nanocrystals can be similarly adjusted in three dimensions by combining them with microcrystals of other optically dense materials. The modular character of the nanoparticle cluster assembly approach thus presents an advantage over direct synthesis of emissive nanostructures, which is often limited to specific materials and conditions. The versatility and the scalability of the approach should lead to improved applications of emissive nanostructures in materials and devices.

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**Supporting Information Available:** Full experimental details and SEM and TEM images of ZnO–CdSe clusters and ZnO rods, emission spectra of TOPO-supported CdSe nanoparticles, and plots of the dependence of the directional emission ratio on cluster length and thickness. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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